Ultrafast X-ray Diffraction Study of Melt-front Dynamics in Polycrystalline Thin Films

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Melting is a fundamental process of matter still not fully understood at the microscopic level. Here we use time-resolved X-ray diffraction to examine the ultrafast melting of polycrystalline gold thin films using an optical laser pump followed by a delayed hard X-ray probe pulse. We observe the formation of an intermediate new diffraction peak, which we attribute to material trapped between the solid and melted states, that forms 50ps after laser excitation and persists beyond 500ps. The peak width grows rapidly for 50ps, then narrows distinctly at longer timescales. We attribute this to a melting band originating from the grain boundaries and propagating into the grains. Our observation of this intermediate state has implications for the use of ultrafast lasers for ablation during pulsed laser deposition.

One Sentence Summary: Laser-induced melting of polycrystalline metals starts at the internal grain boundaries and traps a partially-ordered melting band in the melt front between the solid and melted states as it propagates into the grains.

Introduction
Understanding the role and behavior of transient states during phase transitions, such as melting, is becoming increasingly important in condensed matter physics\((1-4)\). Time-resolved (TR) pump-probe techniques are well-suited to capture these transient states. An optical laser pump pulse induces the phase transition, and an ultrashort X-ray/electron pulse analyzes the properties of the transient state which follows after a specified time delay. Pump-probe methods have been used to study time evolution\((5, 6)\) of thermal and non-thermal melting processes in metals\((7-9)\)
and semiconductors(10, 11). These and other experiments have been widely interpreted with the help of the two-temperature model (TTM)(12), in which the pump pulse is considered to create hot electrons which subsequently transfer their heat to the crystal lattice through electron-phonon coupling within a few picoseconds(13). This model is generally accepted as describing the laser excitation process in a wide range of materials and has been incorporated into a TTM molecular dynamics (TTM-MD) simulation method, which has led to simulations that are found to be in good agreement with experimental results(14–16).

Macroscopic materials are often polycrystalline, containing a large density of differently oriented grains separated by grain boundaries (GBs). Melting has been widely studied over many decades and is believed to be initiated preferentially at defects, such as surfaces, dislocations, GBs, stacking faults, and point defects which contain atoms with lower coordination number thus more dangling bonds than in the bulk(17–19). According to the Lindemann criterion, which is when the amplitude of atomic vibration reaches 10% of the nearest neighbor atomic distance, melting happens at lower temperatures near such defects. Experiments, for example by ion channelling, have shown an increase in disorder at such locations slightly below the bulk melting temperature(20). Theoretical studies have confirmed this concept that melting in granular materials starts to nucleate at dislocations and grain boundaries(21, 22). While the location of the initiation of melting at these weak points of polycrystalline material is widely accepted, the time dependence of the progression of melting is an open question, which has been much less studied. Ultrafast Electron Diffraction (UED) experiments have explored the timescales of melting in free-standing single-crystal films and found different melting regimes(7).
Furthermore, it is well known that the presence of GBs affects electron transport in metals, most easily seen in their electrical conductivity(23). In metal thin films, the electron-phonon coupling constant is related to the inelastic mean free path ($\lambda$) of electrons which varies with energy according to the “Universal curve”(24). In addition, it depends on the electron reflection coefficient from defects such as grain boundaries, film thickness and grain size(25, 26). Depending on the electron energy the inelastic mean free path is in the range of a hundred nm. This implies most of the electrons generated from the front side of a 300nm sample do not make it to the other side, which agrees with experiments(13). In a polycrystalline thin film, however, the separation between GBs, often assumed to scale with the film thickness, can dominate once again and provides a characteristic thickness dependence of resistivity(23). Within the framework of the TTM, it is expected that the hot electrons will couple to the lattice preferentially at GBs because of this additional electron scattering(27, 28). Electrical transport measurements indicate slower effective electron velocities in polycrystalline than in single-crystal gold thin films, due to the additional scattering at GBs(29). This results in a characteristic penetration depth of ultrafast hot electrons which is considerably longer than the 12nm electromagnetic skin depth at 400nm laser wavelength in polycrystalline gold films(13). For both these reasons, we would expect to see heterogeneous melting in a granular metal when it is excited by a laser, with preferential melting at the GBs.

Our experiment was therefore designed to explore the time dependence of laser-induced structural changes in polycrystalline gold (Au) thin films of 50nm, 100nm, and 300nm
thicknesses prepared with the electron-beam evaporation technique. A femtosecond laser pulse was used to excite electrons at the front surface of the Au thin film with a range of fluences spanning the level needed to fully melt the film in a single shot. By measuring the X-ray Diffraction (XRD) patterns of the film with a single X-ray pulse produced by Pohang Accelerator Laboratory X-ray Free Electron Laser Facility (PAL-XFEL), we monitored the structural changes following 400nm optical laser excitation.

**Results**

**Single-shot Time-resolved X-ray Diffraction Results**

The XRD data were collected in a transmission Debye-Scherrer geometry to minimize background shown in Fig. 1(A). A new broader “intermediate” peak is seen to appear on the lower Q side of the (111) powder ring shown in Fig. 1(B). The new powder ring observed here has less azimuthal intensity variations than the (111) powder ring indicating that the contributing material is more homogenous. To show the full Q-dependence, each measured 2D image was azimuthally integrated, to give the XRD profile curves shown in Figs. 2(A), (B), and (C). The powder (111), (200) and (311) Bragg peaks all show two clear peaks, a sharp powder peak, and a broader intermediate diffraction peak at lower Q, which becomes visible about 50ps after laser excitation.

To understand the temporal evolution of the laser-induced changes in the sample, the XRD profiles measured at different times were all fitted with two components, as shown in Fig. 3(A), with full details in the Materials and Methods section below. Both the crystal peak position and intensity, shown in Fig. 3(B) and (C) are seen to decay and oscillate; the oscillation strength
depends weakly on the incident laser fluence. The oscillatory behavior has been explained before to originate from longitudinal acoustic vibration waves traversing the film with a period proportional to the sample thickness\((30, 31)\), while the decay is explained by partial melting\((3, 7)\).

**Origin of the Intermediate Peak**

The intermediate peak position, width, and integrated intensity are well-described by the Gaussian function used in the fitting procedure (details below), as seen in Fig. 3(A). While the intermediate peak position, around \(2.655\text{Å}^{-1}\), is close to the first peak of liquid gold at \(2.64\text{Å}^{-1}\)\((32)\), the peak is ten times sharper than any known liquid gold peak, so cannot be attributed to a conventional equilibrium liquid state. Analogous intermediate peaks are also observed near the (200) and (311) powder peaks (in Figs. 2(B), and (C)) which do not exist at all in the liquid structure factor\((33)\). The peak also cannot be due to the effect of thermal expansion with a continuous distribution of crystal temperatures within the sample because that would give only a monotonic tail on the crystal diffraction peak. Instead, the formation of a distinct peak due to thermal expansion implies there has to be part of the material with a well-defined larger lattice parameter, hence preferred temperature. The only singular temperature of gold is the melting point itself, \(T=T_m\) (1338K): the crystal temperature could be trapped there during heating while it takes up the latent heat of melting. The crystal, undergoing melting, apparently retains a single, well-defined larger lattice constant while it becomes progressively disordered. The splitting of the peak into two distinct components, identified with two different temperatures, is a direct signature of inhomogeneous melting.
The intermediate peak position, shown as a function of time in Fig. 4(C), does not reach the position expected for the lattice parameter of gold at the melting point, shown as a horizontal blue dashed line on Fig. 4(C). We consider that the peak position is offset by the effects of pressure in the melting layer, estimated to be around 5GPa from the bulk modulus and thermal expansion coefficient. The pressure (P) was estimated using, $P = K_0 \Delta V/V$, where $K_0$ is the isothermal bulk modulus for gold (167 GPa) and $\Delta V/V = 3(\Delta Q/Q)$ is the relative volume change determined by the peak shift, $\Delta Q$. Such an induced pressure would be expected to dissipate acoustically, and we note the presence of acoustic oscillations in the peak position, which are particularly strong at 256mJ/cm$^2$ fluence, which also gives the strongest intermediate peak.

**Dynamics of the Crystal Diffraction Peak**

For the 300nm thin film, upon laser excitation, the (111) crystal peak shows an abrupt reduction of intensity for all laser fluences used in our measurements. In addition, the (111) crystal peak position shows a negative position shift due to thermal expansion of the gold. For the data measured at 254mJ/cm$^2$ incident laser fluence, the maximum excursion of the peak is $\Delta Q = -0.0047 \text{Å}^{-1}$ at 60ps after laser excitation, corresponding to a crystal temperature change of 400K, estimated from the thermal expansion coefficient of gold, $14 \times 10^{-6}$ at 300K(34), and assuming no pressure contribution.

In all the 300nm, 100nm, and 50nm thin films, shown in Figs. 5(B), (D) and (E), the sharp decrease of the (111) integrated peak intensity was fitted with an exponential decay function. The dynamics of the crystal peak integrated intensity is different for the different film thickness.
For all the different film thicknesses reported here, the decay time was found to depend strongly on the incident laser fluence. For the 300nm film, measured at fluences of 63, 636 and 2500mJ/cm$^2$, were fitted with exponential decay times of 100±33ps, 85±15ps, and 25±5ps, respectively. These values are close to the observed rise times of the intermediate peak, consistent with a conversion of crystal into the material which gives rise to the new intermediate diffraction peak.

**Analysis of Crystal Peak Oscillations**

Both the integrated peak intensity and peak positions for the 300nm films shows both a decay and damped oscillations. This behavior is interpreted as due to loss of material to melting and acoustic wave oscillations. It is fitted with a function,

$$A(t) = A_0 \left(1 + A_1 e^{-t/\tau_1}\right) + A_2 \left(e^{-t/\tau_2}\right) \cos \left(\frac{2\pi t}{T} + \phi\right)$$

where $A_0$, $A_1$ and $A_2$ are amplitudes, $\tau_1$ and $\tau_2$ are decay times, $T$ is the period of oscillation, and $\phi$ is a phase offset. Both the $\tau_1$ and $\tau_2$ become shorter for higher fluence data; moreover, the time observed here is shorter than the decay time observed in gold nanocrystals(1) which were measured at fluences below the damage threshold. The damping time constant is close to the observed rise time of the intermediate peak, consistent with a conversion of crystal into the material giving the new intermediate diffraction peak. For the data measured at 254mJ/cm$^2$ incident laser fluence, the fit to the peak position gives an oscillation period ($T$) of 136±10ps, corresponds to acoustic waves in a 270±30nm sample thickness propagating at the 3690m/s longitudinal speed of sound in gold films(31). A similar oscillation period was found to fit the integrated peak intensity.
Temperature Rise due to Laser Heating

The pulse energy \( E_{\text{pulse}} \) was calculated by diving the laser power by the 10 Hz repetition rate of the laser. The laser beam size, \( d \), at FWHM was estimated to be 100\( \mu \)m at the sample position (full description is given Materials and Methods section). The incident laser fluence \( (F_{\text{in}}) \) was then taken to be \( E_{\text{pulse}}/A \), where \( A = \pi \left( \frac{d^2}{4} \right) \). \( A \) is the laser beam area. Assuming that the X-rays are probing the central part of the excited region of the sample; we estimate a maximum temperature rise, \( \Delta T \), of the thin film of thickness \( h \), can be estimated as, \( \Delta T = \left( \epsilon F_{\text{in}} \right)/(h\rho C_p) \), where \( \epsilon = 0.6 \) is the absorption coefficient of gold thin at 400 nm\(^{(7)} \), \( \rho \) is the density of gold, 19.3g/cm\(^3\) and \( C_p \) is specific heat capacity of gold, 0.128J/gK. The absorbed fluence \( F_{\text{ab}} \) will be smaller than this reported value due to significant reflection from the surface of the sample. For experiments done on the 300nm film at 127, 190, 254, and 636 mJ/cm\(^2\) incident laser fluences, we estimated the temperature rises of 1030, 1538, 2060, and 5150K respectively, ignoring the latent heat contribution. These estimated temperature values are subject to uncertainties of the probe region. Whenever the laser heating goes into latent heat, \( L = 66J/g \), at the melting point \( T_m = 1338K \), there will be a 516K smaller temperature rise. According to these estimates, 254mJ/cm\(^2\) incident fluence should be just sufficient to melt the 300nm film sample.

Melting Time and Partial Melting

From the estimates given above, we estimate the absorbed fluence for complete melting should be 98mJ/cm\(^2\), 33mJ/cm\(^2\), and 16mJ/cm\(^2\) for the 300nm, 100nm, and 50 nm thin films, respectively. A recent result\(^{(7)} \) showed the threshold for complete melting of 30nm
polycrystalline gold thin films was about 15mJ/cm$^2$, consistent with these numbers. In our data in Fig. 3(C), for the 300nm films, even 636mJ/cm$^2$ incident laser fluence is still not enough to completely melt the film because there is still some intensity left in the crystal powder peak. Both the crystal peak position in Fig. 3(B) and intensity in Fig. 3(C) show an exponential decay and oscillation in the remainder of the partially melted film. At higher fluences, both the peak position and intensity oscillations become weaker. The peak was found to disappear completely within 25ps at 2500mJ/cm$^2$ incident laser fluence, suggesting complete melting of the film at the level of sensitivity of our measurement.

We suppose that this incomplete melting, especially of thicker films, is explained by the limited transmission of the hot electrons through the sample, due to thermal scattering and scattering from the GBs. It was seen in previous work(13) that only a small fraction of the hot electrons are able to traverse a 300nm gold thin film. When the samples are driven with fluences well above the estimates above for complete melting, there must be a significant temperature gradient established by the limited electron transmission: while the far side of the sample is not yet reaching the melting point, the laser-illuminated side goes well above and may reach the boiling point and/or start ablating. This conclusion was tested by examination of the fluence dependence of the decay of the crystal peak intensity of 50nm and 100nm thin films in Figs. 5(B) and (D). For the 100nm thin film, we observe complete melting within 25ps at an incident laser fluence of 440mJ/cm$^2$. Similarly, 20ps is the melting time needed for the 50nm film at 88mJ/cm$^2$ incident laser fluence. Melting becomes faster as the incident laser fluence increases, in good agreement, with the recently reported data on 35nm polycrystalline thin films(7), which were interpreted as
the sample crossing from heterogeneous to homogenous melting. In Fig. 5(F), we plot the residual fraction of crystal left after 500ps of melting against the incident laser fluence, which shows a roughly sigmoid function with a threshold of 153mJ/cm$^2$, for the 300nm film. The 100nm thin film shows similar behavior with a threshold of 60mJ/cm$^2$.

**Discussion**

Having identified that the intermediate peak arises from the material undergoing melting, we can then follow its behavior to report on the structure and properties of the melting region as a function of time and fluence. Since the peak is the structure factor of the melting material, we follow its trends, through lineshape fitting, to understand the overall melting behavior. For the reasons given in the introduction above, we assume that the melting initiates at the GBs of the polycrystalline film where the electrons couple preferentially to the lattice. Heat flow starting from the GBs causes a melting wave, which propagates towards the core of each grain as illustrated schematically in Fig. 6(A).

**Melt Front Velocity**

Classical thermal diffusion calculations can be used to understand the propagation of melting, once the few-ps electron-lattice equilibration time of the TTM has elapsed. This is shown as a temperature-position profile in Fig. 6(B), illustrating how the thermal spike assumed to start at the GBs diffuses rapidly into the neighboring grains and gives rise to a melt front where the temperature crosses the melting temperature, $T_m$. The solution of the 1D heat diffusion equation starting from a point source at the GB where a pulse of heat is injected at time $t=0$, is a spatial
Gaussian distribution, \( T(x, t) \), of width \( \sqrt{2kt/C_p\rho} \) where \( k \) is the thermal conductivity, \( C_p \) is the specific heat and \( \rho \) is the density. According to Fourier’s law, the heat flow, \( \dot{Q} \) or \( (dQ/dt) \), is proportional to the spatial derivative of \( T(x, t) \), which is \( (C_p\rho A/2) T(x, t) (x/t) \), where \( A \) is the area of the GB considered. When \( T = T_m \), the melting temperature, the \( (x/t) \) can be considered as a melt-front velocity, \( v \), determined by the rate of heat flow \( v = 2\dot{Q}/\rho A C_p T_m \). The heat flow, \( \dot{Q} \), is given by the initial conditions and is expected to be proportional to the absorbed laser fluence, which explains why melting times become shorter with fluence, as we and others observe (7).

**Effect of Latent Heat**

Because of the uptake of latent heat, there can be two melt fronts with different velocities \( v_1 \) for the solid-melt boundary and \( v_2 \) for the melt-liquid boundary, indicated in Fig. 6(B). These are given by \( \dot{Q} = (1/2)\rho A v_1 (T_m C_p) = (1/2)\rho A v_2 (L + T_m C_p) \), where \( L \) is the latent heat. This defines a band of melting material sandwiched between the two melt fronts with \( T = T_m \), which is responsible for the intermediate diffraction peak. Assuming a constant heat flow and that both fronts start together, they grow apart with a ratio, \( v_2/v_1 = (L + T_m C_p)/(T_m C_p) = 1.39 \), resulting in enlargement of the melt region.

**Explanation of Narrowing of the Intermediate Peak**

As the melting region propagates through the grain, the gold region undergoing melting between the two melt fronts becomes larger in time, according to this model. While the associated diffraction peak in Fig. 4 shows a rise in both its integrated intensity and width within the first 50
ps, the most significant trend is a distinct narrowing of the peak width over the 100-500 ps delay range in Fig. 4(D). We interpret this to be due to the widening band between the two melt fronts. The physical size of the band, given by $2\pi/W$, where $W$, the full-width at half maximum of the intermediate peak, is 10nm at 100ps growing to 14nm at 550ps for 254mJ/cm$^2$. These values, together with the model above, provide an estimate of the melt front velocity, $v=30$ m/s. Only the integrated intensity shows significant fluence dependence, shown in Fig.4(A), with a drop of intensity for the higher fluence, attributed to faster melting. We interpret the initial 50 ps rise time as the time needed to establish the melt front, while the quasi-stable behavior following this transition corresponds to a steady progression of the melt front while energy is being transferred from the latent heat to the melt.

**Summary and Implications**

Our comprehensive picture of the ultrafast melting of polycrystalline gold films starts when the laser pulse is absorbed within the electromagnetic skin depth of the sample (12nm at 400nm), creating a population of hot electrons which then travel through the sample at Fermi velocity($12, 35$). During this process, hot electrons transfer their energy to the lattice preferentially at the GBs($28$), leading to inhomogeneous melting with a pair of melt-fronts being emitted from each GB, as illustrated in Fig. 6. This melting material gives rise to has a well-defined new diffraction peak with quasi-static width, position, and intensity living beyond 500ps, suggesting that the thermally expanded crystal lattice is partially preserved at $T = T_m$, during the transfer of its latent heat at the melting point. Polycrystalline thin films are prone to have inhomogeneities associated with their grain boundaries, surfaces, dislocations, stacking faults, and point defects.
All these inhomogeneities will have atoms with lower coordination number than in the bulk, leading to spatial inhomogeneities in the electron-phonon coupling rate. For metals, the electron-phonon coupling rate increases with the density of grain boundaries\((23, 24)\). According to\((25)\), increases in the number of grain boundaries leads to a decrease in the mean free path of electrons due to an increase in electron scattering locations. When a femtosecond laser excites a polycrystalline metal thin film, the hot electrons generated will efficiently transfer energy to the lattice at these electron scattering locations. This allows more precise machining of materials with ultrafast lasers\((36)\). It is found that granularity appears when nanosecond lasers are used, so femtosecond lasers are advantageous for precise micromachining which significantly reduces the formation of a “heat-affected zone” in a material\((37)\). Moreover, this will also have a consequence also for laser ablation, perhaps contributing to the formation of particulates found in the ablated plume\((38)\). In both applications, our model may explain why grain averaging when using nanosecond lasers might be needed.

**Materials and Methods**

**Thin film Sample Preparation**

Gold films with a nominal thickness of 50nm, 100nm, and 300nm were fabricated by electron beam evaporation onto standard silicon nitride \((\text{Si}_3\text{N}_4)\) membrane windows at Center for Functional Nanomaterials (CFN), Brookhaven National Laboratory (BNL). The silicon nitride membrane arrays were provided commercially by Silson. The windows are 500×500μm² and 200nm thick in 24x24 or 9x7 arrays on Silicon wafers. Before gold deposition on the membrane, a 2nm Titanium adhesion layer was deposited. To minimize impurities during deposition, the
preparation was done at a pressure of around $10^{-6}$ bar. The sample thicknesses reported here are nominal values, estimated from the deposition rate measured on a quartz crystal balance. We observed no diffraction signal contributions from the Titanium. For the 300nm thin film, the grain sizes were estimated to be $163\pm 40$nm using the Scherrer formula, determined in a separate measurement of the (111) crystal peak width at the Advanced Photon Source, sector 34-ID-C.

**Time-resolved X-ray Diffraction**

We performed time-resolved X-ray diffraction on the polycrystalline gold films at the Pohang Accelerator Laboratory X-ray Free Electron Laser (PAL-XFEL) Facility. The sample was mounted perpendicular to the XFEL beam. The optical pump beam was a few degrees away from normal incidence in an almost co-linear geometry. The pump pulse of 400nm and 100fs was generated from an 800nm Ti: sapphire regenerative amplifier laser system (Coherent, Legend), frequency-doubled with a barium borate (BBO) crystal. The choice of 400nm gives better optical coupling with the polycrystalline gold film than 800 nm since less light is reflected. The laser beam size was estimated by monitoring the laser power transmitted through an aperture at the sample position. With a 200$\mu$m aperture, 90% of the laser power was transmitted. This means the $1/e^2$ beam size, which is often called the laser beam diameter, is less than 200$\mu$m. From this, we estimate the FWHM laser beam size, $d$, is around 100$\mu$m, which is the distance between the 50% intensity points. The incident laser fluence ($F_{\text{in}}$) is then calculated conventionally as pulse energy ($E_{\text{pulse}}$) divided by the area, $E_{\text{pulse}}/A$, where $A=\pi(d^2/4)$. For our experiment, we used a range of incident pulse energies from 5-300$\mu$J, resulting in an incident laser fluence of 63mJ/cm$^2$ to 3.8J/cm$^2$, at the sample position. To probe the uniformly excited part of the sample, the monochromatic XFEL beam was focused to 25$\mu$m (FWHM) spot size at
the sample position with compound refractive lenses. Temporal overlap of both beams was achieved using a GaAs metal-semiconductor-metal (MSM) detector (Hamamatsu) at the sample position, while the spatial overlap was achieved by centering both beams on a 100µm diameter pinhole moved onto the sample position, as seen in Fig. 1(A). The sample was mounted on a motorized scanner which allows for single-shot measurement. The scanner was synchronized to the XFEL beam for "mesh" scans, visiting each window of the array once for each pump-probe delay time. A diffraction image from each shot was collected with a Rayonix MX225-HS area detector, 2×2 binned, at 10Hz. The direct beam intensity was recorded with a quadrant beam position monitor before the sample and on a photodiode after the sample. In order to disregard sample damage, only the first shot on each window was considered in the data analysis. The optical taper geometry of the area detector was pre-calibrated as a fixed correction in the hardware. The data were background subtracted, using both white-field and dark-field corrections, with the latter remeasured once per day. CeO$_2$ powder was used as a calibrant to correct for possible drifts of photon energy and small variations in the sample-to-detector distance between sample changes.

**Data Analysis and Lineshape Fitting**

As a first step, the calibration diffraction image was fit using Fit2D to refine the sample-to-detector distance, detector orientation angles, and center pixel position\(^{(39)}\). Then all the refined values were transferred to PyFAI for azimuthal integration of the measured diffraction images\(^{(40)}\). The direct beam was measured with a photodiode after the sample, and the data were normalized to the incident photon flux recorded at each shot. X-ray diffraction images were
collected at different pump-probe delay times. After calibration, normalization, and integration, each crystalline diffraction peak of the pristine sample was fitted with a sum of two Gaussian functions with constrained intensity ratio and constrained to the same peak position. The peak shape is determined by different factors such as the grain size distribution, the detector resolution, and the beam divergence and is often reported to be Voigt-shaped(41). However, our analysis showed the two-Gaussian peak gave a better fit to the data. The XRD curves for (111) at positive pump-probe delay time showed two distinct peaks: the (111) crystal peak and the neighboring “intermediate” peak. In the peak fitting here, the (111) crystal peak shape was taken to be the same fixed-ratio sum of two Gaussian peaks while the new intermediate peak was modeled with a third single-Gaussian peak. The fitting was done in Python using the lmfit package, which uses non-linear least-squares fitting(42). To limit the fit parameters, the following constraints were introduced:

i) The peak positions of the two Gaussians used for the crystal peak were the same. The widths of the two Gaussian components were fixed at 0.0212Å⁻¹ and 0.063Å⁻¹ (FWHM), optimized by fitting the negative pump-probe delay time data where this is the only peak.

ii) Similarly, the peak intensity ratio between peaks 1 and 2 was fixed to a value 2.47 optimized by fitting the negative pump-probe delay time data.

iii) The intermediate peak was turned on only at the positive pump-probe delay time. The peak position, width, and amplitude were all allowed to be free during the fit.

This lineshape fitting procedure was used to obtain the parameters plotted in Figs. 3, 4, and 5. In addition, we extracted the powder and intermediate peak positions for the different film thicknesses. The intermediate peak position is reported at the time delay, where we observe the
maximum change. For the 300nm film the powder peaks were at $2.6797\AA^{-1}$, $3.096\AA^{-1}$ and $5.125\AA^{-1}$ for the (111), (200) and (311) respectively and the corresponding intermediate peaks were at $2.655\AA^{-1}$, $3.07\AA^{-1}$ and $5.048\AA^{-1}$. Similarly, for the 100nm film, (111), (200) and (311) powder peaks were at $2.693\AA^{-1}$, $3.11\AA^{-1}$ and $5.14\AA^{-1}$ and we were able to resolve only the intermediate peak close to the (111), at $2.648\AA^{-1}$. However, for the 50nm thin film we could only determine the position of (111) and (200) peaks at $2.703\AA^{-1}$ and $3.117\AA^{-1}$ and the intermediate peak position were not resolved. The powder peak positions increase slightly with decreasing thickness, perhaps due to lattice contractions, as reported before(43).

References and Notes


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PJ, ESB, HK, SJLB and IKR performed the data analysis, and the manuscript was written with contributions from all authors; **Competing interests**: The authors declare no competing for financial interests; and **Data availability**: All the data, code, and materials used for the conclusion of this manuscript are available from the corresponding authors, TAA and IKR, upon request.

**Fig. 1. Experimental configuration for time-resolved X-ray diffraction from polycrystalline gold thin films.** (A) Experimental set-up showing the sample, the Rayonix detector **MX225-HS**, and the photodiode \( I_0 \) to measure the transmitted beam for normalization. The sample was mounted perpendicular to the 9.7keV X-ray beam and a 400nm, 100fs optical laser was used to excite the sample in almost co-linear geometry. (B) 2D diffraction patterns of the 300nm thin film collected 50ps before and 100ps, 220ps, and 390ps after laser excitation at 254mJ/cm\(^2\) incident laser fluence. (C) A cross-sectional view of the gold thin-film and substrate window array arrangement.

**Fig. 2. Dynamics of azimuthally integrated XRD profiles of different peaks.** (A) (111) (B) (200) and (C) (311) diffraction peaks for the 300 nm thick, thin film measured at various delay times after 254mJ/cm\(^2\) incident laser fluence excitation. All the diffraction positions have both the crystal powder and intermediate peaks.
Fig. 3. Dynamics of the (111) crystal diffraction peak measured from 300 nm gold thin films. (A) Fits to the diffraction intensity profiles versus momentum transfer, Q, measured 50ps before and 268ps after excitation at 254mJ/cm² incident laser fluence. (B) Peak position and (C) integrated intensity as a function of time delay. For 254mJ/cm² data, the peak position was fitted with a sum of exponential decay (time constant, τ₁=2800±50ps) and an exponentially damped cosine function (with damping time constant, τ₂=90±10ps and period, T= 130±10ps). Similarly, the 254mJ/cm² integrated intensity was fitted with τ₁=2800±100ps, τ₂=250±45ps, and T= 130±10ps.

Fig. 4. Dynamics of the intermediate diffraction peak of the (111) profile. (A) integrated intensity, (B) integrated intensity of the intermediate peak at different incident fluences, (C) position, and (D) width as a function of pump-probe delay time for 300nm films. The data measured at different incident laser fluences are indicated with different colors. The peak width was fitted with an exponential decay convoluted with a Gaussian function. For the data measured at 254mJ/cm² and 127mJ/cm², the time constants were 1320±50ps and 3900±600ps, respectively. The peak position measured at 254mJ/cm² was fitted to a sum of two exponential functions with time constants 50±10ps and 330±20ps convoluted with a Gaussian function. The dashed blue line shown in (C) is the expected position of the gold (111) peak at the melting point due to thermal expansion, assuming ambient pressure.

Fig. 5. Thickness dependence of the melting time of gold thin films. (A) Diffraction intensity profiles measured from a 50nm thick gold film at 50mJ/cm² incident laser fluence at different
delay times after laser excitation. (B) Time dependence of the integrated diffraction intensity of 50nm thin films at different incident laser fluences. (C) Diffraction profiles of a 100nm thick gold film measured at 66mJ/cm² incident laser fluence. (D) The integrated intensity of the (111) peak as a function of pump-probe delay time for different fluences. (E) Time dependence of the crystal peak component for the 300nm gold thin films for different incident laser fluences. (F) Residual crystal fraction versus fluence for 100nm and 300nm thick films. The fraction of the crystal peak intensity remaining 500ps after laser excitation is plotted versus incident laser fluence and fitted with a sigmoid function.

**Fig. 6. Grain boundary (GB) melting mechanism in a polycrystalline gold thin film.** (A) Sketch of GB locations in a polycrystalline gold thin film and a zoomed view of how the melt front would propagate away from the GB following optical laser excitation. (B) Simulation of the spatial temperature distribution using the heat diffusion equation from a spike of melt created at the GB, at x=0. The heat moves rapidly into the grain with a melt front velocity determined by the heat flux. The shaded offset region is due to the uptake of latent heat, resulting in a block of melting material sandwiched between two melt fronts moving at different velocities.
Figure 1.
Figure 2.
Figure 3.
Figure 4.
Figure 5.
Figure 6.